#### 526. The Crystal Structures of Anhydrous Nitrates and Their Complexes. Part I. The a Form of Copper(II) Nitrate

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Copper(II) nitrate in the  $\alpha$  form crystallises with four Cu(NO<sub>3</sub>)<sub>2</sub> in each unit cell of dimensions  $a = 11.12 \pm 0.02$ ,  $b = 5.05 \pm 0.01$ ,  $c = 8.28 \pm 0.02$  Å and space group Pmn2<sub>1</sub>. Each copper atom is co-ordinated by six oxygen atoms from five different nitrate groups in a distorted octahedral [4 + 1 + 1]manner. Each nitrate group forms a bridge between two copper atoms and one of the nitrates forms, in addition, the moderately weak and the weak [1 + 1] bonds in opposite directions from the third oxygen atom. The refinement of the atomic positions following the preliminary report of the structure <sup>1</sup> has thus resulted in a different and more normal co-ordination.

UNTIL the preparation by Addison and Hathaway<sup>2</sup> of anhydrous copper(II) nitrate, it was considered that nitrate groups must be weak ligands. This view was disproved by the chemical stability in anhydrous conditions of copper(II) nitrate; this stability permits it, for instance, to be sublimed without dissociation of the molecules in the vapour phase. The determination of its structure was undertaken to investigate the nature of the bonding in the solid state.

At a late stage in the structure determination, it was realised that solid  $Cu(NO_3)_2$ may be obtained in two crystalline forms, which are very similar in appearance. The preliminary Note <sup>1</sup> and this Paper are both concerned with the  $\alpha$  form, which has only been obtained as a single phase in powder form by heating the compound  $Cu(NO_3)_2 N_2O_4$ in vacuo to about 100°. It is obtained as single crystals by subliming this powder slowly, e.g., in dry air at moderate temperatures, but the sublimate contains a proportion of the  $\beta$  form, which increases on increasing the speed of sublimation. Sublimation at 200° in vacuo produces only the  $\beta$  form, and the  $\alpha$  form is also transformed into the  $\beta$  form on heating (the conversion being rapid above about 150°). The characteristic features in the infrared spectra and in the powder diffraction patterns, by which the two forms may be recognised, have been reported in a separate Note.<sup>3</sup>

The preliminary report <sup>1</sup> of the structure of the  $\alpha$  form emphasised the strong covalent bonding in chains of alternate copper atoms and nitrate groups, but drew what are now found to be incorrect conclusions about the nature of the cross-linking between the chains by further nitrate groups. More recently, an electron diffraction study<sup>4</sup> has shown that distinct  $Cu(NO_3)_2$  units exist in the vapour phase, in which each  $NO_3$  acts as a bidentate ligand giving either a flattened tetrahedral or a square co-ordination of the copper atom with covalent Cu–O bonds of length 2.0 Å making O–Cu–O angles of about 70°.

## EXPERIMENTAL

Anhydrous copper(II) nitrate was prepared as previously described,<sup>2</sup> and crystals of a suitable size for X-ray photography were obtained by sublimation of the powdered material at approximately 180° in dry air at atmospheric pressure in a tube protected from atmospheric moisture by a guard tube containing phosphorus pentoxide. These crystals were either loaded directly into thin Pyrex capillary tubes in a dry box, or first mounted on glass fibres before being placed in the capillaries when alternative crystal orientations were required. In either case, the capillaries were sealed in a small flame within a few seconds of removal from the dry box. Because of the difficulty of handling the crystals, it was necessary to determine X-ray

S. C. Wallwork, Proc. Chem. Soc., 1959, 311.
 C. C. Addison and B. J. Hathaway, Proc. Chem. Soc., 1957, 19; J., 1958, 3099.

 <sup>&</sup>lt;sup>8</sup> C. C. Addison, N. Logan, W. B. Simpson, and S. C. Wallwork, Proc. Chem. Soc., 1964, 341.
 <sup>4</sup> R. E. La Villa and S. H. Bauer, J. Amer. Chem. Soc., 1963, 85, 3597.

intensities on crystals of irregular and inconvenient shapes, for which the calculation of absorption corrections would have been too involved. At first only a- and c-axis oscillation and zerolayer Weissenberg photographs were taken and the corresponding intensities estimated. Later, oscillation and Weissenberg photographs were obtained with the crystals rotating about each of the three crystallographic axes, and intensities for the layer lines 0kl, h0l to h2l, and hk0 to hk4were measured by a photometer method <sup>5</sup> involving the multiple film technique.<sup>6</sup> These intensities were corrected for Lorentz and polarisation effects and converted into  $F_0$  values in the usual way.

The crystal data are as follows:  $\alpha$ -Cu(NO<sub>3</sub>)<sub>2</sub>, M = 187.55, orthorhombic,  $a = 11.12 \pm$  $0.02, b = 5.05 \pm 0.01, s = 8.28 \pm 0.02$  Å, U = 465 Å<sup>3</sup>,  $D_m \simeq 2.5$  (by displacement), Z = 4,  $D_{c} = 2.68, F(000) = 364.$  Space group  $Pmn2_{1}$  (No. 31) or Pmnm (No. 59). Cu- $K_{\alpha}$ -radiation,  $\mu = 66.8 \text{ cm.}^{-1}$ .

### STRUCTURE DETERMINATION

Examination of the Patterson projections on (100) and (001) showed that the copper atoms must be near to the positions  $\frac{1}{4}$   $\frac{3}{4}$ ,  $\frac{$ origin, in either of the two possible space groups, and a number of possible positions for the nitrate groups were also indicated. The 0kl and hk0 structure factors were calculated for a number of possible trial structures with small movements of the copper atoms in various permitted directions from the special positions mentioned. Comparison with the corresponding  $F_{\rm o}$  values (which were the only ones that had been estimated at that time) showed that the space group *Pmnm* did not allow sufficient movement of the copper atoms from the special positions to make it possible to derive any trial structure giving satisfactory agreement between  $F_{\rm o}$  and  $F_{\rm c}$ . This, together with indications of a centric distribution for the hk0 intensities and a non-centric distribution for the 0*hl* intensities, permitted the rejection of the centrosymmetric space group Pmnm. A trial structure having the space group  $Pmn2_1$  and derived from Fourier electron density projections on (100) and (001), phased by the copper atoms only, was refined <sup>1</sup> in these two projections to give R = 0.20. However, attempts to refine this structure further with partial three-dimensional values uncovered difficulties arising from the false symmetry described below. It was therefore decided to make a fresh start with the structure determination using all the intensity data that were available.

Use was made of a programme named "FATAL" newly developed by Hodgson, Rollett and Stonebridge 7 for the MERCURY computer; in this the structure factors are calculated for the input atomic positions and these are used to calculate a three-dimensional Fourier electron density function, which is then scanned for peaks. The output gives the co-ordinates and approximate electron counts of all the peaks above a predetermined size. It also notes the approximate positions of all the peaks that are not locatable by a 19-point routine and gives the interatomic distances between all possible pairs of peaks. Starting only with the copper atom in the position previously determined, most of the peaks found that were above 4 eÅ<sup>-3</sup> corresponded with some of the previous light-atom positions and with related false peaks generated by the two extra approximate mirror planes caused by the nearly special position of each copper atom. Several of these possible peaks could be eliminated as being too close to the copper atom. Other peaks, which were taken to be spurious owing to diffraction effects from the restriction of the intensity data, were in positions related to those of the copper atoms by translations of either a/4 or b/2. The two strongest of the remaining peaks were related by one of the false mirror planes, and each was used in turn as an input atomic position, combined with the copper atom. This use of one position from a pair partly removed the false symmetry, and resulted in two equally acceptable sets of peaks, which could correspond to light atom positions. These peaks still included some related by false symmetry elements, and, in order to find which peaks were genuine, all the possibilities in each set were included in a separate siteoccupation least-squares analysis.<sup>7</sup> In one case, 4 cycles of least-squares reduced R from 0.34 to 0.27 and in the other case 4 cycles reduced R from 0.33 to 0.20. Structure factors and a scan of the Fourier electron-density peaks corresponding to each set of output positions were then obtained by using the "FATAL" programme. The set of peaks obtained from the

<sup>5</sup> S. C. Wallwork and K. J. Standley, *Acta Cryst.*, 1954, 7, 272. <sup>6</sup> J. J. De Lange, J. M. Robertson, and I. Woodward, *Proc. Roy. Soc.*, 1939, *A*, **171**, 398; J. M. Robertson, J. Sci. Instr., 1943, 20, 175.

<sup>7</sup> L. I. Hodgson, J. S. Rollett, and B. R. Stonebridge, to be published.

sites giving R = 0.20 corresponded closely with the atomic positions in the previous partial refinement based on the 0kl, h0l, h1l, hk0, hk1 and hk2 data, with the addition of some extra peaks, which were either too near to the copper atom or too low in electron count to be acceptable. The acceptable peaks were subjected to a normal least-squares refinement,<sup>8</sup> with isotropic temperature factors, using all the available structure factors, with weights  $\sqrt{w} = 1$ if  $|F_0| > 50$ , otherwise  $\sqrt{w} = 50/|F_0|$ . R dropped smoothly in five cycles of refinement to 0.14. The set of peaks obtained from the application of the "FATAL" programme to the sites having R = 0.27 did not correspond to a satisfactory system of interatomic distances. More prolonged attempts at refinement did not reduce R below 0.18 and still resulted in unsatisfactory interatomic distances. This set of possible atomic positions was therefore rejected in favour of the other set corresponding to R = 0.14.

By contrast with the previous attempt to refine the acceptable atomic positions with mainly projection data, the temperature factors in the new least-squares output were quite reasonable. Further refinement, with the more appropriate weighting scheme  $\sqrt{w} = [1 + (|F_0|/20)^2]^{-\frac{1}{2}}$  and with anisotropic temperature factors for all atoms, led to the final set of atomic positions given in Table 1 and to the structure factors of Table 4, which have an agreement index R of 0·117. In view of the rather limited data obtained from crystals of different shapes and dimensions and the lack of correction for absorption, high accuracy in atomic positions cannot be expected and no significance can be attached to the anisotropy of the temperature factors. In Table 1 an equivalent isotropic temperature factor is quoted, and was obtained by taking the mean of the magnitudes of the principal axes of each vibration ellipsoid. The standard deviations of atomic positions derived from the normal equations matrix are also represented by average isotropic values in Table 1.

Τа	BLE	1

Final fractional atomic co-ordinates and isotropically averaged standard deviations and temperature factors

Atom	x/a	у/b	z c	$\sigma_{av}$ (Å)	$B_{\mathbf{av}}$ (Å <sup>2</sup> )
Cu	0.7487	0.7692	0.2469	0.008	1.9
O(1)	0.775	0.295	0.275	0.023	1.4
O(2)	0.801	0.388	0.533	0.024	1.6
O(3)	0.782	-0.011	0.445	0.025	1.4
O(4)	0.902	0.838	0.149	0.025	1.5
O(5)	0.596	0.667	0.347	0.026	1.7
O(6)	0.000	0.655	0.341	0.043	$2 \cdot 5$
O(7)	0.500	0.971	0.203	0.049	$3 \cdot 3$
N(l)	0.784	0.225	0.408	0.028	1.3
N(2)	0.000	0.760	0.210	0.029	0.4
N(3)	0.500	0.784	0.278	0.055	3.0

#### DESCRIPTION AND DISCUSSION OF THE STRUCTURE

The atoms are shown in their final least-squares positions, together with the final Fourier electron-density projections along each of the three axes in Figure 1. A general view of the structure is shown in Figure 2, the viewpoint being the same as that for the corresponding view in the previous note, to show the changes in atomic positions that have taken place subsequently. Although these changes are small for all atoms apart from O(4) and O(6), they lead to a different interpretation of the bonding in the structure. There are still chains parallel to the *a* axis of alternate copper atoms and nitrate groups strongly bonded together, but it is now seen that these chains are also linked sideways by strong bonds between alternate copper atoms and nitrate groups. This sideways linking runs in a zig-zag manner, for example from Cu' near  $\frac{3}{4}$ ,  $\frac{5}{4}$ ,  $\frac{1}{4}$  to Cu near  $\frac{3}{4}$ ,  $\frac{3}{4}$ ,  $\frac{1}{4}$  and back to the repeat of Cu' by the *c* translation, giving rise to corrugated sheets of strongly bound copper atoms and nitrate groups with the average plane of each sheet parallel to (010). The Cu-O bonds within each corrugated sheet form the usual approximately square-planar arrangement, round each copper atom with a mean length of 1.98 Å.

<sup>8</sup> R. Pepinsky, J. M. Robertson, and J. C. Speakman, "Computing Methods and the Phase Problem in X-ray Crystal Analysis," Pergamon Press, Oxford, London, New York, Paris, 1961, p. 107.

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Between these sheets there is weaker Cu–O bonding through O(1). This atom is situated at a distance of 2.68 Å from the copper atom to which the nitrate group, of which it is part, is already linked by the Cu–O(3) bond, but it is also 2.43 Å from the adjacent copper atom in the next sheet, this copper atom being the repeat of the previous one by the b translation. If significant bonding properties are attributed only to the 2.43 Å bond, each copper atom has a co-ordination number of 5, better expressed by the symbol [4 + 1]. If the 2.68 Å distance is also regarded as a bond, then each copper atom has a co-ordination



FIGURE 1. Fourier electron density projections along each of the three crystallographic axes and projections of the atoms in their final least-squares positions. Contours for hk0 and h0l projections (a) and (b) at 5, 10, 20, 30, 50, 70 e.Å<sup>-2</sup>; for 0kl projection (c) at 10, 20, 30, 50, 70, 90, 110 e.Å<sup>-2</sup>

[1965]



Сu

Figure 2. General view of the structure. Broken lines indicate the weaker Cu  $\cdots$  O interactions



FIGURE 3. Bond lengths in (Å) and bond angles

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number of 6, which differs from the more usual [4 + 2] in having unequal weak bonds, and is better expressed as [4 + 1 + 1]. In any case, the nitrate group N(1)O(1)O(2)O(3) forms a bridge between three copper atoms and, if the copper atom co-ordination is taken as being [4 + 1 + 1], then the extra weak bond Cu-O(1) together with the strong bond Cu-O(3) gives some bidentate character to the bonding of this nitrate group. This is thought to be the first example of this type of bonding by a nitrate group, and it emphasises the strong ligand properties of this group. The other nitrate groups in the structure perform the more usual function of bridging pairs of copper atoms.

The details of the environment of each copper atom and the bond distances and most of the bond angles are given in Figure 3. The bond angles that are not illustrated in this figure are given in Table 2. Average standard deviations for bonds between light atoms

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IAI	SLE 2		
Further O-Cu-	O angles (degr	rees)	
	Cu-O(1)		Cu-O(1'')
Cu-O(1'')	163		0
Cu-O(2)'	74		120
Cu-O(3'')	116		50
Cu-O(4)	96		76
Cu-O(5)	79		109
O(2') - Cu - O(3'')		169	
O(4)-Cu-O(5)		175	

and for Cu–O bonds are 0.04 and 0.03 Å, respectively Since these may be underestimates, in view of the nature of the initial data, it is not appropriate to discuss the bonding



in detail. However, there is no doubt that the four strong Cu-O bonds with an average length of 1.98 Å can be distinguished from the two weaker ones, and that the latter are unequal in length. Also, as might be expected, the N-O bonds to oxygen atoms that are strongly bound to copper are consistently longer than those to oxygen atoms that are not strongly bound, the average lengths for the two types of bond being 1.30 and 1.16 Å, respectively.

It is possible, by means of the following argument, to gain some insight into the way in which the structure arises. The maximum co-ordination numbers of copper atoms and nitrate groups can be taken as 6 and 3, respectively, and, if these were realised, the correct stoicheiometric ratio would be obtained. However, if the nitrate group is too bulky to permit the attachment of six different nitrate groups to each copper atom, the next-best choice is to have each copper atom attached to five different nitrate groups by allowing half the nitrates to form three Cu–O bonds and half to form only two such bonds. This is the actual situation in the structure of  $\alpha$ -Cu(NO<sub>3</sub>)<sub>2</sub>. The layers perpendicular to the *a* axis consist of a pseudo-hexagonal array of copper atoms and nitrate groups each with a co-ordination number of 3 within the layer. The linking into chains parallel to the *a* axis through nitrate groups each with co-ordination number 2 then increases the copper co-ordination number to five.

The simplest way of building up the pseudo-hexagonal layers is shown in an idealised manner in Figure 4(a). This involves a trigonal arrangement round each copper atom, which would not be consistent with the usual tendency to octahedral co-ordination for copper atoms without bending the bonds so much out of the plane that the oxygen atoms would be brought too close together. Two possible arrangements involving linear O-Cu-O bonds are shown in Figures 4(b) and (c). In (b) an *anti*-arrangement is maintained for each pair of nitrate groups, and it can be seen that this leads to a cyclic structure that is too open in character and only permits a two-fold co-ordination within the layer for each copper atom. The corresponding structure with each pair of nitrates in a syn-arrangement is shown in Figure 4(c). This allows each copper atom to be co-ordinated by three different nitrate groups and gives a closely packed structure. Slight rotations of the nitrate groups in the directions indicated by arrows allow each copper atom to form a fourth weak bond within the layer as indicated by the dotted lines. This is the structure actually adopted in the  $\alpha$  form of anhydrous Cu(NO<sub>3</sub>)<sub>2</sub>, as can be seen by comparing the unit cell indicated by broken lines in Figure 4(c) with Figure 1(c). This comparison also shows that the bridging nitrate groups above and below the layer take up positions such that the oxygen atoms that are bonded to each copper atom are as remote as possible from the oxygen atoms in the layer. Nevertheless, there are a number of oxygen-oxygen contacts of less than 3.0 Å, as listed in Table 3. It is presumably the repulsive forces resulting when this crystal

TABLE 3

(	Oxygen al	toms attache	ed to the same copp	Oxygen atoms attached to different copper						
O(1) · · ·	O(2')	2·71 Å	$O(2') \cdots O(5)$	2.86 Å	$O(3) \cdots O(4; 1\frac{1}{2} - x, 1 - y, \frac{1}{2} + z) 2.79 A$					
$O(1^{\prime\prime}) \cdot \cdot$	$\cdot O(4)$	2.90	$O(3'') \cdots O(4)$	2.90	.,					
O(1) · · ·	$\cdot O(5)$	$2 \cdot 80$	$O(3^{\prime\prime}) \cdots O(5)$	2.75						
$O(2') \cdot \cdot$	• O(4)	2.71								

$O \cdots O$ contacts less	than $3.0$ Å	between different	nitrate groups

structure is formed (in order to satisfy the bonding requirements of both the copper atoms and the nitrate groups), which cause its instability. The structure in the vapour phase 4 has the lower co-ordination numbers of 4 and 2 for the copper atoms and nitrate groups respectively. The repulsion is relieved, but the Cu-O-N angle of about 85° is less favourable than the average of 114° in the crystal structure of the  $\alpha$  form. It is presumably for this reason that nitrate ligands are rarely bidentate in the solid state and usually perform the function of bridging groups. A determination of the crystal structure of  $\beta$ -Cu(NO<sub>3</sub>)<sub>2</sub> is in progress.

TABLE 4

Observ	ved s	truct	ure amj	plitudes	com	pare	d with	final ca	lculat	ted st	tructure	e ampli	tudes	and	phases
<b>hki</b> 002	$ F_0  = 109$	$ F_c $ 130	α (deg.) 195·7	hkl 224	$ F_0 $ 59	Fc  57	α (deg.) 3·8	<i>hkl</i> 503	$ F_0 $ 29	$ F_c $ 24	α (deg.) 24·7	<i>hkl</i> 814	$ F_0 $	$ F_c $ 10	α (deg.) 357·3
004 006	86 56	75 47	353-7 175-8	225 226	18 49	14 50	256.7 174.9	507 509	11 17	9 15	205·4 311·8	815 816	31 6	29 7	$354.2 \\ 132.1$
008 00-10	34 35	$\frac{32}{33}$	16·4 166·6	$227 \\ 228$	$15 \\ 35$	$\frac{13}{35}$	341·0 12·9	$510 \\ 511$	46 24	$\frac{32}{22}$	90-0 264-8	817 818	$^{28}_{5}$	29 6	$167.5 \\ 54.5$
010 011	20 97	$\frac{26}{142}$	0·0 244·9	229 22·10	4 12	5 14	$152.7 \\ 189.9$	$512 \\ 513$	$\frac{22}{19}$	$19 \\ 20$	169·6 87·4	820 821	58 19	55 17	180-0 319-4
$012 \\ 013$	18 70	15 70	125·8 178·4	$230 \\ 231$	10 34	8 36	0·0 352·3	$514 \\ 515$	$\frac{10}{25}$	$\frac{11}{28}$	72·7 72·7	822 823	52 8	49 8	358·7 168·3
014 015	61 40	50 34	283-2 341-6	232 233	24 48	26 52	$191.7 \\ 167.5$	516 517	17	18	315-8 50-5	824 825	41 17	41 17	179·3 347·1
016	$\frac{1}{27}$	25 27	227.1 131.6	240 241	52 28	49 28	180-0 351-5	518 519	25 4	28 4	172·8 23·2	826 828	29 23	29 23	354.5
019 01-10	39 3	41 5	338·0 194·2	$\overline{242}$ $243$	40 33	37 29	26·8 201·8	$520 \\ 521$	19 17	16 17	90-0 276-1	830 831	22 34	16 40	180·0 172·9
020	89 23	$120 \\ 26$	180-0 342-6	244 250	25 25	23 23	162·9 180·0	522 523	$\frac{1}{22}$	19 17	295-2 160-1	833 840	28 36	30 36	359-8
022 023	89 60	$113 \\ 59$	5.5 130.7	251 253	$\frac{1}{24}$	21 21	186.6	524 525	31 13	29 12	285-2 151-9	842 844	30 26	31 31	166-8 355-8
024 025	55 16	50 15	192·8 42·6	254 260	10	$12 \\ 17$	137·1 0.0	526 527	14	$12 \\ 10$	63·2 21·8	851 852	22 20	22 18	356-0 177-8
026 027	49 24	45 25	349·4 214·8	261 262	$16 \\ 20$	$15 \\ 16$	171.5 185.7	528 529	7	8	3·4 67·2	853 901	$19 \\ 23$	18 15	160·5 137·2
028	34 7	38	186.9	263 301	22 48	23	183·2 331·9	531 532	26 23	26 18	124·3 159·6	903 907	24 12	18	131.3
02·10 030	12 18	$15 \\ 13$	9.0 0.0	303 305	36 16	28 11	4·7 316·9	540 543	14 18	12 16	270-0 115-2	910 912	19 17	16 20	90.0 341.4
031 032	46 37	44 36	176-6 302-7	307 309	28 12	23 10	150·8 42·4	544 553	$12 \\ 10$	12 11	98.5 144.0	913 914	12 11	14	104·4 139·1
033	53 16	65 16	346·7	310 311	57 24	44 27	270-0 267-8	600 602	106	84 63	180-0	916 917	7	11	284.9
035 037	27 31	30 35	175·1 333·3	$\frac{312}{313}$	$\frac{20}{17}$	18 14	180-9 261-7	604 606	45 41	36 39	166.6	918 920	8 13	9 11	165-5 270-0
039 040	18 51	18 53	145.4	314 315	 9 12	7	329·5 227·3	608 610	35 11	33 12	181.7 180.0	921 925	11	10	230-5
041 042	36 53	30 46	146·2 192·1	316 317	14 11	14 11	5·2 260·4	611 612	53 4	57	177·5 214·1	926 927	56	4 4	227·6 89·3
043 044	40 49	33 47	354·1 329·3	319 31.10	4	6 4	224·8 348·6	613 615	$37 \\ 26$	43 24	346·0 173·7	930 10-02	16 44	10 37	270-0 351-5
045 046	22 34	22 43	196·9 179·2	320 321	37 7	40 7	90-0 293-2	616 617	6 29	6 28	56·2 349·8	10-04 10-06	49 38	41 37	174·8 2·4
048 050	21 34	$17 \\ 32$	340-6 0-0	323 324	18 18	$15 \\ 16$	4·2 164·5	619 620	24 65	26 66	165-3 0-0	10·10 10·11	6 24	6 22	180.0 197.9
$   \begin{array}{r}     051 \\     053   \end{array} $	37 47	22 37	352-2 159-7	325 326	11 10	9 9	188-6 109-3	621 622	$19 \\ 57$	$\begin{array}{c} 21 \\ 60 \end{array}$	143·4 176·7	$10.12 \\ 10.13$	5 34	7 35	82-0 350-4
054 055	$\frac{22}{33}$	$\frac{22}{31}$	321·8 351·3	327 328	14 5	$^{14}_{5}$	335·1 177·2	623 624	$\frac{12}{39}$	$\begin{array}{c} 12 \\ 40 \end{array}$	324·8 359·5	$10.14 \\ 10.15$	14 37	15 37	284·6 175·6
056 060	$17 \\ 12$	$16 \\ 13$	164·4 180·0	329 330	$10 \\ 12$	11 10	183·5 90·0	625 626	9 32	$\frac{11}{31}$	$193 \cdot 2 \\ 176 \cdot 2$	10·16 10·17	7 35	8 30	264·6 0·5
$\begin{array}{c} 061 \\ 062 \end{array}$	$\frac{25}{18}$	$\frac{24}{19}$	350·6 347·0	333 334	11 11	$\frac{14}{16}$	31·9 348·5	627 628	8 26	8 28	63·6 3·7	$10.20 \\ 10.21$	$\frac{29}{11}$	$\frac{26}{10}$	0-0 198-4
063 064	22 17	23 20	$     183 \cdot 2 \\     172 \cdot 2   $	343 344	$\begin{array}{c} 13 \\ 12 \end{array}$	14 9	$228.9 \\ 19.3$	629 630	8 20	9 20	197·5 0·0	$10.22 \\ 10.23$	33 22	28 18	$169.1 \\ 80.7$
101	10 42	5 33	73·3 161·9	400 402	167 95	146 94	0.0 193.3	$\begin{array}{c} 631 \\ 632 \end{array}$	33 23	35 20	350-8 134-4	10.24 10.25	38 10	38 10	$354.9 \\ 162.6$
107	12	6	353-9 90-0	404 406	84 65	97 66	359-1 176-7	633 640	32 29	34 29	179-9 180-0	10.26	24 21	26 20	180-1 0-0
112	33 22	33 19	87.5	408 40·10	30 16	25 16	173.1	642 643	23 17	24 20	12.9 160.7	10.31	26 16	29 18	355-3 189-3
114	25 5	22 4	37-1	410	33 57	24 62	349·6	644 653	27 18	30 16	188.2	10.40	26 20	23 21	180-0 347-5
119	21 21	23	87·3 270·0	412 413	60 60	16 78	251.0 175.7	654 701	7 34	9 29	134.9 328.3	11.01	20 18	17 16	334-0 206-7
121 122 102	11	21 9	229·0 344·8	414	17 53	18 60	301·3 350·1	703	30 15	20 12	348·4 292·0	11.05	10	12 12	278·1 270·0
125	23 8 15	18	14·2 86·3	416	34	34 7	178.0	707	28 11	24 12	40.1	11.12	9 8 5	10	56.9 190.5
125	7	6	279.9	419	23	23	230·1 330·1	711	42	18	287.5	11.20 11.92	10	8	255·5 270·0
127 12.10 191	3	4	78·5	41.10	65 02	70	173.1	712 713	20 14	19	276-0	11.23 11.24 11.95	13	10	99-8 99-8
133	11	19	230.7 341.0	421 422 492	23 73	25 80	3.9	714	8	10	222.5	12.00	36 52	35	203·5 0·0 172.1
200	145	133	180.0	424	59 18	60 17	184.5	717	10	11	254.3	12.02 12.04 12.10	29	24	359.0
204 206	83 78	92 73	175-2	426	48 12	49 12	351.1	720	35 13	28 13	90·0 358·5	12.11 12.12	39 10	47	9.6 126.4
208 20-10	37 26	35 25	206-0 352-8	428	29	32 45	182.9	724	13	11	161·2 200·3	12.13 12.14	28	28	179-2
210 211	11 37	8 42	180-0 118-7	432	17 47	18	32.7	726	9 14	8 14	101.5	12.15 12.20	23 35	19 34	343.6
$     \frac{1}{212}     213 $	40 54	38 60	102·1 348·1	440 441	60 21	61 21	0.0	728 730	4 17	5 15	170-9	$12 \cdot 21$ 12 · 22	13 29	10 28	311·5 355·5
214 215	19 59	15 60	67·1 172·7	442 443	46 29	43 31	181·4 23·2	743 750	11 14	16	237·2 270·0	12.23 12.24	8 30	-7 28	$132 \cdot 4$ 175 \cdot 3
$\frac{216}{217}$	16 36	16 31	52·5 338·0	444 450	33 24	$34 \\ 25$	333-8 0-0	800 802	85 88	71 78	0.0	$12.30 \\ 12.31$	14 28	12 28	180-0 182-0
$218 \\ 219$	12 32	$12 \\ 32$	75·3 161·9	451 453	$\overline{34}$ 29	33 27	353-1 155-4	804 806	51 36	43 34	350-1 182-2	$13.12 \\ 13.14$	4 2	-7 2	291·0 96·2
$21 \cdot 10 \\ 220$	6 56	6 70	347·2 0·0	454 460	$\frac{12}{27}$	$16 \\ 23$	336·3 180·0	808 810	$\frac{24}{22}$	27 17	337-8 0-0	$13.20 \\ 14.00$	6 18	6 13	90-0 180-0
$\begin{array}{c} 221 \\ 222 \end{array}$	34 72	35 86	227.7 178.1	$461 \\ 462$	20 23	21 22	354·1 347·0	811 812	59 9	64 9	5.9 112.0	14.02 14.11	19 20	26 28	339·7 187·6
223	<b>32</b>	30	19-6	501	77	77	166-4	813	44	50	173-2				

The authors thank Professor C. C. Addison for suggesting the problem and for continued interest and discussions, Dr. N. Logan for assistance in the preparation and mounting of the crystals, Mrs. A. M. B. Douglas, J. S. Rollett, and B. R. Stonebridge for considerable assistance with the computing, D. W. J. Cruickshank, J. S. Rollett, L. I. Hodgson, and R. A. Sparks for the use of computer programmes, the Directors of the Computing Laboratories at the Universities of Leeds and Oxford for computing facilities, and Mrs. D. H. Thomas for long and patient assistance with the desk-machine calculations and in the preparation of diagrams.

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[Received, August 10th, 1964.]